THE EFFECT OF PRE-STRAIN ON FATIGUE FOR A HIGH STRENGTH LOW ALLOY (HSLA) STEEL


The effect of initial plastic strain cycle fatigue lives of HSLA specimens is investigated. Specimens were monotonically, quasi-statically loaded under strain control in tension to produce plastic strains (prestrain conditions $s_0 = 0$, $s_1 = 4.6$, $s_2 = 35.6$ and $s_3 = 66.7\%$ deformation). Surface analysis on the annealed Zn coat on the prestrain surfaces shows that, the higher the deformation the bigger and deeper the cracks on the Zn coat. Results show that the higher prestrain the higher fatigue resistance for all stress applied, specimens with $s_3$ shows the highest fatigue resistance for these experimental conditions.

Keywords: HSLA steel, fatigue, predeformation, stamping

INTRODUCTION

During the stamping process sheet steels are prestrain at different levels that have an effect on the fatigue strength of the component, some previous research have shown that the prestrain on this conditions need to be consider in the structural designs. It has been found that HSLA steels get hard during fatigue test, and fatigue strength decreases as deformation increases decreasing its ductility, on the other hand for lower deformation amplitudes there is a lower plastic deformation on a prestrain material increasing its fatigue strength [1]. Fatigue behavior of prestrain and as-received is different for IF and DP steels [2], some studies shown that there is an effect of fatigue strength for different prestrain levels in HSLA decreasing fatigue strength as prestrain increases for high cycles above $10^5$ [3].

Other study on a DP600 steels shows that fatigue strength decreases as prestrain increases at cycles below $10^4$ but fatigue strength increases for cycles above $10^4$. On the other hand a study in Advanced High-Strength Steels showed that the DP steels exhibit a continuous yielding behavior, low yield point, and a high strain-hardening coefficient [4,5] and for TRIP steels, the main parameters that can affect the monotonic deformation are retained austenite and its carbon content, morphology, and distribution [6]. TRIP 780 showed an improved strain life compared with a DP 590, which may be related to a combination of higher yield strength, higher ductility, and crack tip retardation by austenite to martensite transformation. Both steels exhibited cyclic softening, similar levels of softening are believed to be due to similar ferrite volume fractions of the two steels, as well as similar low-strain work-hardening rates [7].

Researches examining the fatigue properties of several steel grades of similar strength have shown endurance limits increasing in the order of high-strength low-alloy HSLA, DP, and TRIP [8,9]. These results indicate that while the fatigue performance of steel is often linked to monotonic tensile strength, the microstructure may have some effect on the endurance limit. The aim of this work is to measure fatigue strength for a HSLA steel at cycles $< 10^5$ under different prestrain conditions.

EXPERIMENTAL PROCEDURE

An HSLA steel with a thickness of 1.4 mm was employed in this work, chemical composition was obtained by spark spectroscopy as shown in Table 1.

Optical and electronic microscopies were employed to measure microstructure as received and after fatigue tests, in order to analyzed grain size and crack formations. Sample preparation and grain size were made according ASTM E3 and ASTM E112 respectively.

Tensile test were made in order to characterize HSLA steel mechanically, tests were made in a Universal Shimadzu Autograph AG-X, 300 kN capacity. Tensile specimens were machined and made according ASTM E8, geometry and dimensions of the specimen are shown in Figure 1, in order to determine the maximum level of uniform elongation for the as-received material before fatigue testing. Four different prestrain levels were introduce to the specimens, prestrain conditions $s_0 = 0$, $s_1 = 4.6$, $s_2 = 35.6$ and $s_3 = 66.7\%$ deformation using uniaxial stretching, these deformation values were $s_1$ yield point, $s_3$ maximum tensile strength and $s_2$ a point in between.
control in tension loading at room temperature, fatigue tests were stopped if failure existed or if specimen did not fail after 200,000 cycles.

RESULTS AND DISCUSSION

Microstructure of HSLA steel obtained results in a ferrite and cementite phases where grain boundary and lamination direction are observed, grain size results in $6.3 \pm 0.3 \ \mu m$ Figure 2 a, on the surface a Zn coat was observed.

Three tensile tests were made parallel to the lamination direction for the same sample, Figure 3 a, in order to obtain strain-hardening exponent $n$ and the strength coefficient $K$ only a portion between 10% and 20% of true strain is taken to plot a log $\sigma$ vs log $\varepsilon$, plastic zone, Figure 3 b. The value of the slope is the strain-hardening $n$, extrapolation of log $\varepsilon = 0$ is the value of the strength coefficient $K$.

Results show a maximum stress at $464.7 \pm 1.4 \ \text{MPa}$, a yield stress of $350.85 \pm 0.24 \ \text{MPa}$ an elongation percentage of $26.47 \pm 1.18 \ %$; on the other hand in the plastic region shows a value of $n = 0.21 \pm 0.003$ show-
D. AGUIRRE-GUERRERO et al.: THE EFFECT OF PRE-STRAIN ON FATIGUE FOR A HIGH STRENGTH LOW ALLOY STEEL

ing that neck formation will starts at 21% deformation and $K = 776.21$ MPa, with a value of $r = 1.21 \pm 0.13$ showing that this alloy has a degree of anisotropy due to its grain orientation.

Surface analysis was made after prestrain in order to see its condition before fatigue tests, crack formations were observed, the higher the deformation the bigger and deeper the cracks, most of the cracks are perpendicular to the strain direction, Figure 4, transversal micrographs shows that crack formation and propagations were observed only on the galvannealed Zn coating. For a galvannealed coating the alloying reaction results in a hard, relatively brittle coating, in this case the coating is around 200 microns and crack are observed only on this surface.

Fatigue test

Fatigue test results are shown in Figure 5, where at higher stress values fatigue life decreases and at lower stress values fatigue life increases as expected; at the lowest stress 279.5 MPa specimens did not fail and fatigue test was stopped after 200 000 cycles.

Results show that the higher prestrain the higher fatigue resistance for all stress applied, specimens with $s_0$ and $s_1$ seem to behave very similar under the different experimental conditions as $s_2$ prestrain is in the limit of the plastic rage. On the other hand $s_3$ shows the highest fatigue resistance for all stresses applied, these behavior is explain in terms of work hardening as $s_3$ and $s_2$ conditions were prestrain above the plastic limit.

Finally in all cases a transgranular fracture was observed having a higher deformation close to the fracture surface, see Figure 6.
CONCLUSIONS

A fatigue test behavior was studied for an HSLA steel, ferrite and perlite microstructure was observed. Tension test was evaluated at 90° from the laminating direction having the maximum stress at 465 MPa, a strain hardening \( n = 0.21 \pm 0.003 \) and strength coefficient \( K = 776.21 \).

Results show that the higher prestrain the higher fatigue resistance at the stresses studied, finding a transgranular fracture.

Acknowledgments

Authors would like to thanks to PEMSA and CONACYT Mexico, for its founding support.

REFERENCES


Note: The responsible for english language is the lector from Faculty of Mechanical and Electrical Engineering, Nuevo Leon, Mexico